

# Large Post-growth Energy Band-gap Tuning of The 980 nm High Power Laser Diode Structures

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**Abstract**—Post-growth energy band gap turning of the 980nm high power semiconductor laser structure through the quantum well intermixing (QWI) has been investigated. The QWI was carried out by depositing a thin film of SiO<sub>2</sub> on top surface of the laser structure samples and followed by high temperature annealing. By using the QWI technique, band gap energy of the 980nm quantum well structure has been blue shifted up to >220nm. High quality of the laser diode structure after the QWI has been confirmed by fabricating the high performance semiconductor lasers using the wafer after the QWI.

## 1.Introduction

Post-growth energy band gap tuning of the quantum well structures has very important application in developing novel photonic devices. Selective area quantum well intermixing (QWI)[1] has been reported as the post-growth band-gap energy tuning technique for developing photonic integrated circuits (PICs) and quantum well infrared photo-detectors (QWIPs). Comparing to other techniques, QWI is very attractive for the post-growth energy band gap tuning the semiconductor nanostructures because of its simplicity and effectiveness. Semiconductor nanostructures, e.g.

quantum wells and quantum dots, have been largely used for developing high performance novel electronic and optoelectronic devices. Post-growth energy band gap tuning of the AlInGaAs/InP quantum wells has been reported with their photoluminescence (PL) wavelength blue shift of 110nm after the QWI[2]; by capping with a Si<sub>3</sub>N<sub>4</sub> layer and annealing at 800 °C, the blue shift of an InGaAs/InP QW reaches 145nm[3]; The quantum wells/dots intermixing for the post-growth bandgap energy tunings have been intensively investigated[4-9], and very large energy band gap tuning has been received from the InGaAsP/InGaAs/GaAs material system based QW and QD structures.

Different intermixing techniques for the post-growth band gap energy tuning of the semiconductor QW structures and QD structures have been developed[8]. Among the different intermixing techniques, quantum well intermixing by depositing a dielectric capping layer on top of the samples followed by rapid thermal annealing (RTA) is very attractive since it does not require sacrificial layers and avoids damaging the sample-surface by the high energy ion beams bombardment with other intermixing techniques.

In this research, the post-growth band gap energy tuning of a 980nm InGaAs/InGaAsP single quantum well

high power laser diode structure through the QWI has been studied. The QW intermixing was carried out by using a SiO<sub>2</sub> layer to cap the sample and followed by rapid thermal annealing (RTA) at high temperature. A remarkably large blue shift up to 220nm of the QW laser structure has been achieved after the QW intermixing. This is the largest blue shift of 980nm QW laser structure through QW intermixing reported so far, which can be used for developing a multi-wavelength high power laser chip.

## 2. Experiment

The samples used in this study were In<sub>0.24</sub>Ga<sub>0.76</sub>As<sub>0.52</sub>P<sub>0.48</sub>/In<sub>0.24</sub>Ga<sub>0.76</sub>As/GaAs single quantum well structures for a 980nm high power laser diode structure. Table 1 summarizes the layer structure of the samples used in this study.

Table 1 epi-layer structure of the 980nm InGaAs/InGaAsP SQW for QWI

Material	Thickness (nm)	Doping cm <sup>-3</sup>	Remark
Ga <sub>0.76</sub> In <sub>0.24</sub> As <sub>0.52</sub> P <sub>0.48</sub>	100	undoped	waveguide
Ga <sub>0.76</sub> In <sub>0.24</sub> As	8	undoped	active layer
Ga <sub>0.76</sub> In <sub>0.24</sub> As <sub>0.52</sub> P <sub>0.48</sub>	182	undoped	waveguide
n-Ga <sub>0.51</sub> In <sub>0.49</sub> P	1500	~5×10 <sup>17</sup>	Cladding
n-GaAs	500	~5×10 <sup>17</sup>	Buffer
n-GaAs substrate		>1×10 <sup>18</sup>	Substrate

The QW structure was grown on the (100)-oriented GaAs (Si-doped: ~1×10<sup>18</sup>cm<sup>-3</sup>) substrates by metal organic vapor phase epitaxy (MOVPE). Before growing the QW structure, a 500nm GaAs (Si-doped:~5×10<sup>17</sup>cm<sup>-3</sup>) buffer layer was grown at 720°C to suppress the surface defect of the substrate, and then, a 1500nm n-type Ga<sub>0.51</sub>In<sub>0.49</sub>P cladding layer (Si-doped:~5 × 10<sup>17</sup>cm<sup>-3</sup>) was grown at 680°C. After that, a 182nm undoped Ga<sub>0.76</sub>In<sub>0.24</sub>As<sub>0.52</sub>P<sub>0.48</sub> lower waveguide layer was grown at 700°C, followed by

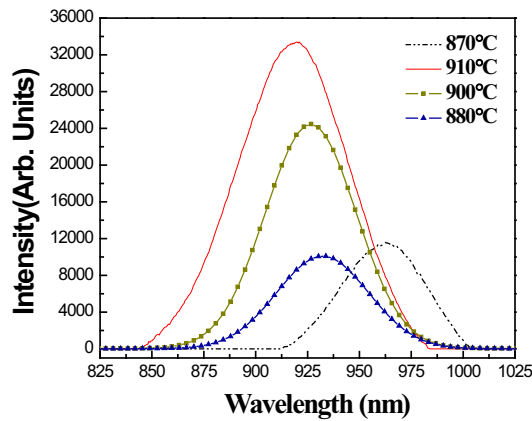
growing the 8nm Ga<sub>0.76</sub>In<sub>0.24</sub>As single quantum well layer. It was grown at 710°C. Finally, the 100 nm Ga<sub>0.76</sub>In<sub>0.24</sub>As<sub>0.52</sub>P<sub>0.48</sub> upper waveguide layer was grown at 700°C.

For carrying out the QWI for the sample, the SiO<sub>2</sub> thin layers were deposited on top of the sample by the ion plasma sputtering, respectively. During the SiO<sub>2</sub> sputtering deposition, the Ar gas flow was set at 28sccm, the plasma power was 180W, the substrate temperature was set at 100 °C and the deposition time was 44mins. After the SiO<sub>2</sub> deposition, the samples were then thermal annealed by an RTA at different temperatures for the QWI. The annealing time was fixed at 60s, which is shorter enough to minimize the surface decomposition of the sample during the QWI.

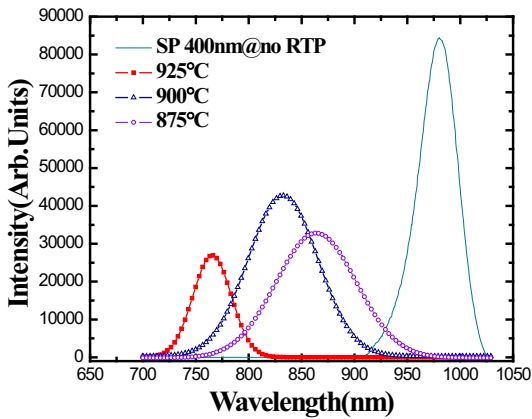
To measure the band-gap energy changes of the quantum well structure after the QWI, the blue shift of the photoluminescence (PL) emission of the quantum well structure after the QW intermixing has been measured and investigated. To investigate the effect of the QWI, different SiO<sub>2</sub> cap layer thicknesses and annealed at different temperatures for the QWI have been investigated. The PL measurements were carried out at room temperature by using an argon ion gas laser with a wavelength of 475 nm as the excitation source. The laser excitation power was kept the same for all the measurements. The emission signals from the samples were dispersed by a monochromator and detected by a Si detector.

## 3. Results and Discussion

Figure 1 shows the measured photoluminescence spectra of the samples after the QWI with different annealing



(a)



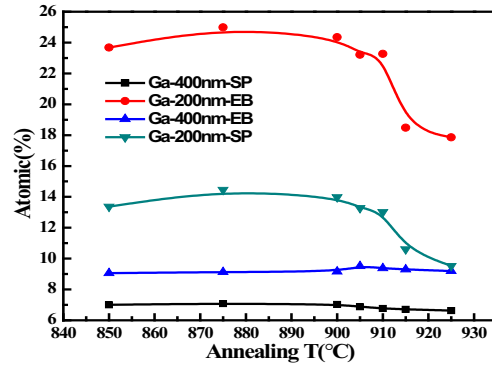
(b)

Fig.1. The measured photoluminescence (PL) spectra of the samples capped with different SiO<sub>2</sub> thickness (a) 200nm, (b) 400nm after intermixing at different annealing temperatures

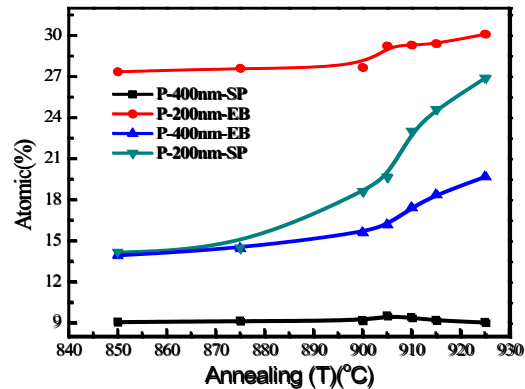
temperatures. The SiO<sub>2</sub> cap layers with different thicknesses for the QWI were deposited by using the Ar plasma sputtering. Sample (a) was capped with 200 nm SiO<sub>2</sub> cap layer, while SiO<sub>2</sub> cap layer of samples (b) was 400 nm thick.

The shortest emission wavelength of sample (a) after the intermixing was measured at 919nm when the annealing temperature was at 880°C. The maximum blue shift of the emission wavelength of the samples after the intermixing was about 70nm. When the SiO<sub>2</sub> cap layer thickness was increased

to 400nm, the shortest emission wavelength of the sample after the intermixing reached 760nm. The maximum blue shift of the sample after the intermixing reached 220nm after annealed at 925°C.



(a)



(b)

Fig. 2. The measured atomic percentage (gallium (a) and Phosphorus(b) of the sample by energy dispersive X-Ray spectroscopy (EDX) after the intermixing

Figure 2 shows the measured percentage of Ga element on top surface of the samples after the QWI. Fig.3(a) shows the changes of the Ga atoms on the samples' surface after the QWI with different annealing temperatures. It also compares the measured Ga atoms of the samples capped with 200 nm and 400 nm

SiO<sub>2</sub> layers deposited by using E-beam and sputtering, respectively.

The out-diffusion of gallium element of the sample after the intermixing will reduce the blue shift of the QW, while the out-diffusion of the arsenic element will increase the QW's blue shift. Out-diffusion of P element makes the samples blue shift stronger.

Figure 2 also shows the suppression of gallium and phosphorus atoms' diffusion with different thicknesses and different SiO<sub>2</sub> deposition methods. The different stabilities of Ga and P at different annealing temperature is the main reason that cause the PL wavelength blue shift.

It is well-known that sputter-deposited SiO<sub>2</sub> trends to be polycrystalline. The polycrystalline SiO<sub>2</sub> thin film has stronger suppression of the surface out-diffusion of the sample than that of amorphous SiO<sub>2</sub>. At the same time, phosphorus element is much more volatile during the annealing process at high temperatures, especially, above the annealing temperature of 900°C. The thickness of the SiO<sub>2</sub> cap layer inevitably affects the out/in-diffusion of the phosphorus element. Increasing the SiO<sub>2</sub> thickness and reducing the quantity of pores will also increase the in-diffusion phosphorus, further blue-shifting the PL peaks.

808nm high power semiconductor lasers were fabricated from the samples with and without the QWI.

Figure 3 shows measured emission spectrum of the high power laser diode device fabricated from the wafer without QWI with the injection current above the threshold current. The device was measured lased at 808nm wavelength.

The device fabricated from the wafer after gone through the QWI was measured lasing at 792nm with injection current of 1.0A at room temperature.

The QWI was carried out with the sample's SiO<sub>2</sub> cap layer of 500nm by sputtering and annealed at 800 °C.

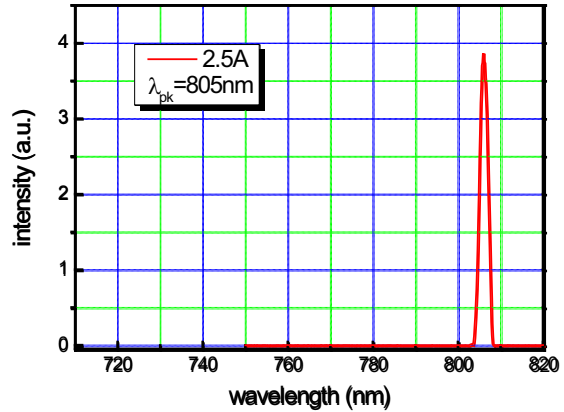


Figure 3 The measured emission spectrum of the high power laser diode fabricated from the wafer without QWI.

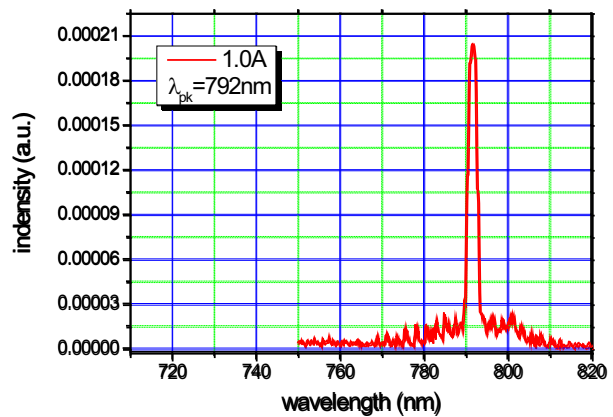


Figure 4 The measured emission spectrum of the high power laser diode fabricated from the wafer after the QWI.

It shows that lasing of the device fabricated from the wafer after the QWI has been received. The device performs well. But the lasing wavelength blue shift of the device with the QWI only about 16nm, which is limited. Further investigation is necessary.

#### 4. Conclusion

Post growth energy band gap tuning of InGaAs/InGaAsP single quantum well 980nm high power laser structure through quantum intermixing has been investigated. The quantum well intermixing was carried out by depositing a thin layer of SiO<sub>2</sub> on top surface of the quantum well structure samples and followed by high temperature RTA annealing. Large blue shift of the sample's emission wavelength has been observed after the QWI with the 400 nm sputter deposited SiO<sub>2</sub> layer samples, the maximum blue shift of the quantum well structure's emission wavelength reaches up to 220nm after the intermixing. The blue shift of the sample after the intermixing was reduced when the SiO<sub>2</sub> cap layer thickness was reduced to 200nm.

Lasing has been received from the sample after the QWI and the emission wavelength was observed blueshifted.

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